PCT

WORLD INTELLECTUAL PROPERTY ORGANIZATION International Bureau



INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

INTERNATIONAL APPLICATION FUBLISH	_	SNDER THE PATENT COOPERATION TREATT (FCT)
(51) International Patent Classification ⁵ :		(11) International Publication Number: WO 94/21726
C08L 23/04, C08J 5/18, B29C 55/22	A2	(43) International Publication Date: 29 September 1994 (29.09.94)
(21) International Application Number: PCT/EP (22) International Filing Date: 17 March 1994 (patent (AT, BE, CH, DE, DK, ES, FR, GB, GR, IE, IT, LU,
(30) Priority Data: MI93A000553 23 March 1993 (23.03.93)	1	Published Without international search report and to be republished upon receipt of that report.
(71) Applicant (for all designated States except US): SOTE [IT/IT]; Via San Martino, 30/32, I-20017 Rho (IT)		L.
(72) Inventor; and (75) Inventor/Applicant (for US only): LANDONI, Guido Corso Europa, 197, I-20017 Rho (IT).	rl\ttl] c	7:
(74) Agent: GERVASI, Gemma; Notarbartolo & Gerva Viale Bianca Maria, 33, I-20122 Milano (IT).	asi S.r.	1.,
(54) Title: SINGLE-LAYER BIAXIALLY ORIENTED H ETHYLENE AND RELEVANT PRODUCTION		HRINKABLE FILM COMPRISING LINEAR LOW DENSITY POLY- CEDURE
(57) Abstract		
Single-layer biaxially oriented heat-shrinkable film cobiaxial orientation of a mixture of ethylene/ α -olefin linear and optionally a crystallization inhibitor.	emprisir copoly	ng linear low density polyethylene obtained by extrusion and twin-bubble mer, a polymeric alloy and/or an ethylene-propylene random copolymer
,		
·		

FOR THE PURPOSES OF INFORMATION ONLY

Codes used to identify States party to the PCT on the front pages of pamphlets publishing international applications under the PCT.

ΑT	Austria	GB	United Kingdom	MIR	Mauritania
ΑÜ	Australia	GE	Georgia	MW	Malawi
BB	Barbados	GN	Guinea	NE	Niger
BE	Belgium	GR	Greece	NL	Netherlands
BF	Burkina Faso	HU	Hungary	NO	Norway
BG	Bulgaria	DE DE	Ireland	NZ	New Zealand
BJ	Benin	<u></u>	Italy	PL	Poland
BR	Brazil	JP	Japan	PT	Portugal
BY	Belarus	KE	Ketiya	RO	Romania
CA	Canada	KG	Kyrgystan	RU	Russian Federation
CF	Central African Republic	KP	Democratic People's Republic	SD	Sudan
CG	Congo		of Korea	SE	Sweden
CH	Switzerland	KR	Republic of Korea	SI	Slovenia
CI	Côte d'Ivoire	KZ	Kazakhstan	SK	Slovakia
CM	Cameroon	Li	Liechtenstein	SN	Senegal
CN	China	LK	Sri Lanka	TD	Chad
cs	Czechoslovakia	LU	Luxembourg	TG	Togo
CZ	Czech Republic	LV	Latvia	TJ	Tajikistan
DE	Germany	MC	Monaco	TT	Trinidad and Tobago
DK	Denmark	MD	Republic of Moldova	UA	Ukraine
ES	Spain	MG	Madagascar ,	US	United States of America
Fī	Finland	ML	Mali	UZ	Uzbekistan
FR	France	MN	Mongolia	VN	Viet Nam
GA	Gabon		-		

PCT/EP94/00852

10

20

SINGLE-LAYER BIAXIALLY ORIENTED HEAT-SHRINKABLE FILM COMPRISING LINEAR LOW DENSITY POLYETHYLENE AND RELEVANT PRODUCTION PROCEDURE

Prior art

The use of linear medium and low density polyethylene for the production of single-layer biaxially oriented heat-shrinkable film has been known for a long time.

Linear polyethylene has found extensive application in the packaging sector as heat-shrinkable films, which require a good transparency and gloss, high mechanical resistance combined

Generally, however, there are various reasons for the biaxial orientation of linear polyethylene being harder than that of normal branched polyethylene.

First of all, orientation is hindered by the polymer structure being free from long branched chains and having a higher crystallinity percentage.

Furthermore, the density being the same, linear polyethylene - compared with branched polyethylene - exhibits the following characteristics:

- a higher melting point by 10°C to 20°C approx.
- just a few very short side chains

with high shrinking values.

- narrow molecular weight distribution.

Collectively, the above features hinder orientation.

25 However, the experimentation on linear polyethylene was a

15

20

source for the development of particular types of such polyethylene suitable for orientation.

For example, USP 4,497,920 (Du Pont) claims the production of a heat-shrinkable film consisting of an ethylene/α-olefin linear copolymer having two distinct crystalline melting points below 128°C.

The existence of two melting areas according to the above patent allows an easy biaxial orientation of the film, which can be processed between the two melting points.

As known, polymers like linear polyethylene which tend to crystallize on cooling generally exhibit one crystalline melting point by calorimetric analysis.

It is, therefore, very hard to determine the temperature or a temperature range at which the extruded tubular film is sufficiently ductile for blowing (with biaxial orientation) and at the same time sufficiently tough to resist blowing pressure. It follows that the twin-bubble orientation of crystallizable polymers (linear polyethylene, polypropylene and polyolefins in general) is extremely critical, in particular in the case of a single-layer film.

Summary

It has surprisingly been found that a single-layer heatshrinkable film comprising linear low density polyethylene can be obtained by a procedure comprising the steps of:

a) preparing a mixture consisting of:

PCT/EP94/00852

WO 94/21726

3

- an ethylene/α-olefin linear copolymer,
- a polymeric alloy and/or an ethylene-propylene random copolymer, and optionally
- a crystallization inhibitor;
- b) extruding, in the melted state, the mixture obtained under 5 a);
 - c) rapid cooling of the tubular film leaving the extrusion die to a temperature of 28°C to 32°C and subsequent heating to a temperature close to softening;
- d) biaxially orienting the film by the twin-bubble technique; 10
 - e) rapid cooling of the biaxially oriented film to 25°C-35°C.

The claimed procedure allows an easy and precise control of the operating parameters as well as the obtainment of films with improved mechanical, optical and heat seal resistance

properties. 15

25

Detailed description of the invention

The present invention contemplates a single-layer biaxially oriented heat-shrinkable film comprising linear low density polyethylene and relevant production procedure.

20 The primary component of said film is an ethylene/a-olefin linear copolymer with 4-8% by wt. a-olefin.

Said copolymer is added with 10 to 40% by wt. heterophasic polymers alloy composed of a matrix consisting of propylene homopolymer and a disperse phase consisting of a synthetic elastomer, e.g. ethylene/propylene.

4

The ethylene/ α -olefin linear copolymer may be alternatively or also added with 10 to 50% by wt. ethylene-propylene random copolymer. A non-alternative additive may be an oligomer of the so-called crystallinity inhibitors group.

The addition, if any, of said oligomer in a by wt. % amount ranging from 1 to 20 in respect of the basic copolymer depends on the type of primary mixture (more or less rich in crystallizable polymeric components), on the procedure and on the operating conditions (more or less favouring crystals formation), and to the desired final characteristics of the product.

In particular, the expression "linear ethylene/ α -olefin copolymer" means a copolymer of ethylene and from 4 to 8% by wt. butene or octene or α -methyl pentene, having linear molecular chains without branchings and without cross-links. The preferred α -olefin is octene.

The copolymer density ranges from 0.900 to 0.935 $\rm g/cm^3$ and melt index from 0.7 to 4.0.

The expression "polymeric alloy" - also known as "catalloy" 20 means a product obtained by catalytic reaction composed of a
matrix consisting of propylene homopolymer and a disperse phase
(copolymer).

In particular, the disperse phase of said alloys may consist of α -olefins of the C_2 - C_3 or C_3 - C_4 type.

Density is 0.88 to 0.90 g/cm^3 and melt index is 0.7 to 4.

PCT/EP94/00852

20

The expression "ethylene-propylene random copolymer" means a synthetic product obtained from ethylene and propylene, the latter being the major component, e.g. 10% ethylene and 90% propylene.

- Out of the oligomers inhibiting crystallization, the following products are suitable for the purpose:
 - aliphatic and aromatic hydrocarbon resins, aliphatic and aromatic copolymers, such as piperylene, methylbutene, vinyltoluene, indene, a-methylstyrene, polycyclodiene, etc.,
- hydrogenated C₉ resins, styrene, a-methylstyrene, isobutene, pinene and rosin resins, and terpene resins.
 - The ethylene/ α -olefin linear copolymer can be hardly oriented by the twin-bubble procedure.
- In fact, on approaching the crystalline melting point, the polymer loses consistency and the bubble obtained by tubular film inflation is extremely unstable.
 - We have found that the addition of ethylene-propylene polymeric alloy and/or of ethylene-propylene random copolymer yields a mixture that can be easily handled thanks to its wide temperature range within which the tubular film is sufficiently ductile to be blown and sufficiently tough to resist blowing pressure.
 - A further possibility of mixture adjustment consists in the addition of crystallization inhibitory oligomers.
- The present invention introduces some innovations in the

WO 94/21726 PCT/EP94/00852

6

5

10

15

20

25

traditional twin-bubble procedure, which allow an accurate product control during the process critical phases.

According to the claimed procedure, proper proportions of the single components are mixed in a conventional slow mixer with separate mixture preparation or by an automatic mixing and metering apparatus on the extruder hopper.

Extrusion is carried out by single-screw extruders, 28-32 diameters long.

The melted mixture leaves the extruder die in a tubular form at a temperature of 180°C to 205°C. The tubular film diameter is a function of the desired max. stretching ratio of the final film.

As concerns polyolefins, it is usual practice to use a balanced stretching ratio in the two directions (longitudinal and transversal), which ratio may range from 4 to 6. The stretching ratio used in the claimed procedure is 1:4.5 approx.

The tubular film leaving the extruder is sized and cooled by a new method in respect of the conventional twin-bubble extrusion technique. In particular, the tubular film, which is still hot, is sized by passing between an outer ring, wherefrom a liquid coolant at 15°C-25°C flows continuously, and a mould, internal to the tubular film, which is also cooled by a liquid coolant at 15°C-25°C.

The max. temperature variation along the tubular film circumference is 1°C. A more uniform thickness and a greater

15

underneath.

homogeneity of the extruded material are thus secured.

By internal and external cooling the tubular film temperature drops to 28°-32°C in few seconds, preferably within 4 to 9 sec. The water leaving the outer ring adheres, as a liquid film, to the tubular film outer surface and flows to the chamber

This provides an accurate control of the rate of crystal formation in the product.

The cold tubular film is then fed to an IR-ray or hot-air oven

where it is heated to a temperature close to softening. Once
heated to the desired temperature, the tubular film is inflated
with compressed air and expanded in the transversal direction.

This results in a transversal orientation of the molecules.

The tubular film inflation is carried out by controlling the operating temperature within ±1°C.

At the same time, the tubular film is stretched in the longitudinal direction by causing the pull-roll upstream of the oven and the final pull-roll to operate at a different speed.

Once the film has undergone double hot stretching, it is rapidly cooled to maintain the inner orientation of molecules, resulting from transversal inflation and longitudinal stretching.

To this purpose, the stretched film is cooled to 25°C-35°C by blowing air at 5°C to 15°C within 15 to 30 sec.

WO 94/21726 PCT/EP94/00852

8

The film obtained according to the claimed procedure is 10 to $40~\mu m$ thick and exhibits improved characteristics over the product obtained by the known technique.

In particular, the film of the invention having thickness between 12 and 25 μm exhibits:

- haze: from 1.0 to 2.0% depending on thickness
- surface gloss at 20°C: from 100 to 120 units
- tensile strength: from 100 to 120 N/mm2
- shrinking at 120°C: from 50 to 60%, balanced in the two directions
 - heat seal resistance: >60 N/mm2

5

15

20

- tearing propagation: from 10 to 25 g depending on thickness.

 Furthermore, the procedure under the invention gives a film with characteristics tailored to the various uses.
- Thanks to the above characteristics, the film as per the present invention is suitable for the packaging sector in general and in particular for the applications requiring a good film shrinkage and adhesion to the packed product as well as a good mechanical and heat seal resistance. For example, the claimed film is particularly fit for the packaging of vegetables, boxes, food also of irregular shape, books and magazines, multiple packed products, etc.

The following examples are conveyed by way of illustration, not of limitation.

10

15

20

EXAMPLE 1

A single-layer film, 15 μm thick, consisting of the following raw materials was produced.

The basic polymer used was linear polyethylene containing 8 % by wt. octene (melt index 1 and sp. gr. 0.923), in particular Clear Flex FG 308 produced by Enichem Polimeri.

Said polymer was added with 20% by wt. propylene-ethylene polymeric alloy (melt index 0.8 and density 0.89 g/cm³), in particular the elastomeric copolymer Hi Fax 7023 XEP produced by HIMONT.

The crystallinity inhibitor was a completely saturated hydrogenated hydrocarbon resin, Eastotac 140 produced by Eastman Kodak, having molecular weight of 900 approx. and Brookfield viscosity of 1000 approx. Said resin, after dispersion in low density polyethylene (50%) to give the masterbatch (Prisma AD WAX 90237 produced by Frilvam) was added to the basic polymer mixture in a 5% amount equivalent to 2.5% pure resin.

Finally, antiblock and sliding additives were used to improve film handling in production and application phases. In particular, 0.5% micronized silicon dioxide was used as antiblock agent (equivalent to 1% masterbatch of type AB 6004 produced by Constab).

The operating conditions were as follows:

25 - temperature of the melted material leaving the die: 181°C

PCT/EP94/00852

- cooling water: 19°C ± 1
- cooling temperature of tubular film leaving the die: 30°C
- cooling time of tubular film leaving the die: 5 sec
- diameter of tubular film leaving the die: 180 mm
- 5 longitudinal/transversal stretching ratio: 4.5
 - tubular film blowing temperature: 120°C
 - cooling temperature of stretched tubular film: 30°C
 - cooling time of stretched tubular film: 20 sec.

Table 1 shows the characteristics of the film obtained and, for the purpose of comparison, the characteristics of commercial product CLYSAR LLP (DuPont).

The film obtained according to Example 1 showed an improved tear strength in respect of CLYSAR LLP.

EXAMPLE 2

20

The procedure of Example 1 was repeated with the difference that the mixture of raw materials was further added with ethylene-propylene random copolymer (melt index 1.8), in particular with 10% Moplen EP2 S 34 F (HIMONT).

The type and quantity of hydrocarbon resin and of antiblock and sliding additives remained unchanged.

The product obtained showed higher elastic modulus and improved optical characteristics. It was also stiffer than that as per Example 1 and, therefore, more adequate for use in automatic fast packaging machines.

25 Also handling conditions improved, the bubble stability being

higher.

The melted material temperature ranged from 185°C to 188°C and the tubular film blowing temperature was 126°C.

Table 1 shows the characteristics of the film obtained.

5 Compared with the product as per Example 1, the film had a higher heat seal resistance and improved optical properties.

EXAMPLE 3

The procedure of Example 1 was repeated with the difference that the mixture was further added with ethylene-propylene random copolymer (melt index 4), in particular with 10% Moplen SYL 7008 XCP (HIMONT).

The type and quantity of hydrocarbon resin and of antiblock and sliding additives remained unchanged.

The product obtained exhibited improved heat sealing characteristics (lower sealing temperature in respect of the product as per Example 2, which means shorter sealing times and wider sealing range). Also seal resistance was improved.

Furthermore, like in Example 2, the temperature range was wider than in Example 1, and bubble stability was the same.

20 Operating parameters:

melted product temperature: 185°C

tubular film blowing temperature: 124°C

Table 1 shows the characteristics of the film obtained.

Compared with the products as per the above mentioned Examples.

25 heat-sealed film had a high tear strength.

EXAMPLE 4

The procedure of Example 1 was repeated with the difference that the polymeric alloy was completely removed and the quantity of ethylene-propylene random copolymer (melt index 4),

in particular Moplen SYL 7008 XCP (HIMONT) was increased (25%).

The type and quantity of hydrocarbon resin and of antiblock and sliding additives remained unchanged.

The product obtained had excellent heat seal characteristics.

On handling, bubble stability and thickness distribution were excellent.

Operating parameters:

melted product temperature: 182°C

bubble blowing temperature: 122°C

 15 Table 1 shows the characteristics of the film obtained.

The mechanical properties were worse than in the previous cases.

EXAMPLE 5

The procedure of Example 1 was repeated with the difference that the hydrocarbon resin was added in a 1.5% amount in respect of the basic copolymer.

The type and quantity of antiblock and sliding additives remained unchanged.

The product obtained was a medium-stiff and medium-shrinkable film.

The operating conditions were similar to those of Example 1: temperature of melted product leaving the die: 183°C tubular film blowing temperature: 123°C

Table 1 shows the characteristics of the film obtained.

5 Compared with the previous products, this film exhibited good mechanical properties and excellent tear strength.

EXAMPLE 6

The procedure of Example 1 was repeated with the difference that the basic copolymer was linear polyethylene (melt index 1.1; sp. gr. 0.920) containing 8% octene, with more uniform

The operating conditions were analogous to those of Example 1 (additional 5°C on extrusion).

The film characteristics were very similar to those of Clysar LLP, as shown in Table 1.

distribution (Dowlex NG 5056 E produced by Dow Chemical).

EXAMPLE 7 (Comparison)

The procedure of Example 2 was repeated with the difference that the basic resin was added with 7% propylene-ethylene polymeric alloy and with 2% random copolymer.

All mechanical properties were worse (Table 1).

EXAMPLE 8 (Comparison)

The procedure of Example 1 was repeated without addition of polymeric alloy.

The bubble could not be maintained at stable and uniform conditions and frequent bursts occurred. No improvement even by

WO 94/21726 PCT/EP94/00852

14

considerably varying all operating conditions. Significant and measurable samples could not be obtained.

TABLE 1: Film characteristics

. .	Ex.1	Ex.2	Ex.3	Ex.4	Ex.5	Ex.6		Commercial CLYSAR LLP (Du Pont)
Thickness µ	15	15	15	15	15	15	15	15
Tensile strength N/mm ²	120	130	110	95	125	100	85	100
Elongation at break %		115	130	125	130	130	90	130
Heat seal resistance g/cm	950	110 0	1200	1000	1050	1100	800	1100
Tear strength g	20	15	22	18	24	16	10	10
Shrinking at 120°C %	56	55	55	55	53	52	52	50
Haze %	1.7	1.4	1.4	1.2	1.4	1	1.3	1
Gloss (photoelec	tric ce 100	ll uni: 105	ts) 105	105	100	110	100	110
Elastic modulus N/mm ²	348	360	340	300	352	315	350	300

CLAIMS

- 1. Procedure for the preparation of a single-layer biaxially
- 2 oriented heat-shrinkable film comprising linear low density
- 3 polyethylene comprising the steps of:
- a) preparing a mixture consisting of:
- 5 an ethylene/α-olefin linear copolymer,
- a polymeric alloy and/or an ethylene-propylene random
- 7 copolymer, and optionally
- 8 a crystallization inhibitor;
- 9 b) extruding, in the melted state, the mixture obtained under
- 10 a);
- c) rapid cooling of the tubular film leaving the extrusion die
- 12 to 28°C to 32°C and subsequent heating to a temperature close
- 13 to softening;
- 14 d) biaxially orienting the film by the twin-bubble technique;
- e) rapid cooling of the bioriented film to 25°C-35°C.
- 1 2. The procedure according to claim 1 wherein said ethylene/ α -
- 2 olefin linear copolymer contains from 4 to 8% by wt. α-olefin
- 3 selected out of the group consisting of butene, octene, hexene,
- 4 and α-methylpentene.
- The procedure according to claim 1 wherein said ethylene/α-
- 2 olefin linear copolymer has density ranging from 0.900 to 0.935
- g/cm3 and melt index ranging from 0.7 to 4.0.
- 1 4. The procedure according to claim 1 wherein said polymeric

PCT/EP94/00852 WO 94/21726

alloy is a product obtained by catalytic reaction, composed of 2

17

- a matrix consisting of a propylene homopolymer and a disperse 3
- phase consisting of C_2 - C_3 α -olefins. 4
- 5. The procedure according to claim 1 wherein said ethylene-1
- propylene random copolymer is obtained by the synthesis of 2
- ethylene and propylene, the latter being the major component. 3
- 6. The procedure according to claim 1 wherein said 1
- crystallization inhibitor is selected out of the group 2
- consisting of aliphatic and aromatic hydrocarbon resins, 3
- 4 piperylene. methylbutene. vinyltoluene. indene.
- a methylstyrene, styrene, pinene and rosin resins, and terpene 5
- 6 resins.
- 7. The procedure according to claim 1 wherein said mixture
- 2 prepared under step a) contains 10 to 40% by wt. of said
- 3 polymeric alloy in respect of said ethylene/α-olefin linear
- copolymer and/or 10 to 50% of said ethylene-propylene random 4
- copolymer and, if necessary, 1 to 20% of said crystallization 5
- inhibitor. 6
- 1 8. The procedure according to claim 1 wherein said cooling of
- step c) causes the tubular film temperature to drop to 28°C to 2
- 32°C in 4-9 seconds.
- 9. The procedure according to claim 1 wherein said biaxial 1
- orientation is obtained at a longitudinal/transversal
- stretching ratio of 1:4.5.
- 10. The procedure according to claim 1 wherein said cooling of

WO 94/21726 PCT/EP94/00852

18

- step e) causes the tubular film temperature to drop to 25°C to
- 35°C in 15-30 seconds.
- 1 11. Single-layer biaxially oriented heat-shrinkable film
- 2 made of composition comprising:
- 3 a linear low density polyethylene,
- 4 an ethylene/ α -olefin linear copolymer with 4-8% by wt.
- 5 α-olefin,
- 6 a polymeric alloy and/or an ethylene-propylene random
- 7 copolymer, and optionally
- 8 a crystallization inhibitor.

PCT

WORLD INTELLECTUAL PROPERTY ORGANIZATION International Bureau



INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

(22) International Filing Date: 17 March 1994 (17.03.94) (30) Priority Data: MI93A000553 23 March 1993 (23.03.93) IT	ated States: AU, BR, CA, CN, IP, KR, US, Europea tot (AT, BE, CH, DE, DK, ES, FR, GB, GR, IE, IT, LI , NL, PT, SE).
MI93A000553 23 March 1993 (23.03.93) IT Mit in (88) Veröffe (71) Applicant (for all designated States except US): SOTEN S.R.L. [IT/IT]; Via San Martino, 30/32, I-20017 Rho (IT). (72) Inventor; and (75) Inventor/Applicant (for US only): LANDONI, Guido [IT/IT]; Corso Europa, 197, I-20017 Rho (IT). (74) Agent: GERVASI, Gemma; Notarbartolo & Gervasi S.r.l.,	
(CA) TEAL. CONCLE LANCE BLAVIAL I V ODUDITED HEAT CHERIKADLE I	uernationalem Recherchenbericht. entlichungsdatum des internationalen Recherchen

(54) Title: SINGLE-LAYER BIAXIALLY ORIENTED HEAT-SHRINKABLE FILM COMPRISING LINEAR LOW DENSITY POLY-ETHYLENE AND RELEVANT PRODUCTION PROCEDURE

(57) Abstract

Single-layer biaxially oriented heat-shrinkable film comprising linear low density polyethylene obtained by extrusion and twin-bubble biaxial orientation of a mixture of ethylene/ α -olefin linear copolymer, a polymeric alloy and/or an ethylene-propylene random copolymer and optionally a crystallization inhibitor.

FOR THE PURPOSES OF INFORMATION ONLY

Codes used to identify States party to the PCT on the front pages of pamphlets publishing international applications under the PCT.

AT	Austria	GB	United Kingdom	MR	Mauritania
AU	Australia	GE	Georgia	MW	Malawi
BB	Barbados	GN	Guinca	NE	Niger
BE	Belgium	CR	Greece	NL	Netherlands
BF	Burkina Faso	HU	Hungary	NO	Norway
BC	Bulgaria	IE	Ireland	NZ	New Zealand
BJ	Benin	IΤ	Italy	PL	Poland
BR	Brazil	JР	Japan	PT	Portugal
BY	Belarus	KE	Kenya	RO	Romania
CA	Canada	KG	Kyrgystan	RU	Russian Federation
CF	Central African Republic	KP	Democratic People's Republic	SD	Sudan
CG	Congo		of Korea	SE	Swcden
CH	Switzerland	KR	Republic of Korea	SI	Slovenia
Ċl	Côte d'Ivoire	KZ	Kazakhstan	SK	Slovakia
CM	Cameroon	LI	Licchtenstein	SN	Senegal
CN	China	LK	Sri Lanka	TD	Chad
CS	Czechoslovakia	LU	Luxembourg	TG	Togo
CZ	Czech Republic	LV	Latvia	T.J	Tajikistan
DE	Germany	MC	Monaco	11	Trinidad and Tobago
DK	Denmark	MD	Republic of Moldova	UA	Ukraine
ES	Spain	MG	Madagascar	US	United States of America
FI	Finland	ML	Mali	U2	Uzhekistan
FR	France	MN	Mongolia	VN	Viet Nam
GA	Gabon		<u>-</u>		

Intel mal Application No PCT/EP 94/00852

A. CLASSIFICATION OF SUBJECT MATTER IPC 5 C08L23/04 C08J5/18 B29C55/22 According to International Patent Classification (IPC) or to both national classification and IPC **B. FIELDS SEARCHED** Minimum documentation searched (classification system followed by classification symbols) CO8L CO8J B29C IPC 5 Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched Electronic data base consulted during the international search (name of data base and, where practical, search terms used) C. DOCUMENTS CONSIDERED TO BE RELEVANT Relevant to claim No. Citation of document, with indication, where appropriate, of the relevant passages Category * 1-11 Y EP, A, O 434 322 (E.I.DU PONT DE NEMOURS) 26 June 1991 see page 3, line 49 - page 4, line 13; claim 6 1-11 Y EP.A.O 423 387 (HEISEI POLYMER CO., LTD) 24 April 1991 *whole document* EP,A,O 519 837 (ECP-ENICHEM POLYMERES 1-11 A FRANCE S.A.) 23 December 1992 see claims A EP,A,O 321 220 (MITSUI PETROCHEMICAL 1-11 INDUSTRIES, LTD) 21 June 1989 see page 2, line 50 - page 3, line 17; claims -/--Further documents are listed in the continuation of box C. Patent family members are listed in annex. * Special categories of cited documents: "T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the 'A' document defining the general state of the art which is not considered to be of particular relevance invention "E" earlier document but published on or after the international "X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to filing date document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified) involve an inventive step when the document is taken alone document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such docu-"O" document referring to an oral disclosure, use, exhibition or ments, such combination being obvious to a person skilled in the art. other means document published prior to the international filing date but later than the priority date claimed "&" document member of the same patent family Date of the actual completion of the international search Date of mailing of the international search report 0 9. 09. 94 29 August 1994 Name and mailing address of the ISA Authorized officer European Patent Office, P.B. 5818 Patentiaan 2 Tel. (+31-70) 340-2040, Tx. 31 651 epo nl, Fax: (+31-70) 340-3016 Clemente Garcia, R

INTERNATIONAL SEARCH REPORT

Inte: anal Application No
PCT/EP 94/00852

C.(Continua	ion) DOCUMENTS CONSIDERED TO BE RELEVANT	
Category *	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
A	EP,A,O 369 658 (SUMITOMO CHEMICAL COMPANY) 23 May 1990 see claims	6
		·
:		

INTERNATIONAL SEARCH REPORT

information on patent family members

Inter mal Application No
PCT/EP 94/00852

Patent document cited in search report	Publication date	Patent family member(s)	Publication date
EP-A-0434322	26-06-91	NONE	
EP-A-0423387	24-04-91	NONE	
EP-A-0519837	23-12-92	FR-A- 267798 CA-A- 209000 WO-A- 930040 JP-T- 650104	7 22-12-92 1 07-01-93
EP-A-0321220	21 - 06-89	JP-A- 115784 DE-A- 387622 US-A- 527798	3 07-01-93
EP-A-0369658	23-05-90	JP-A- 213524 CA-A- 200123 US-A- 503264 US-A- 511502	4 16-05-90 5 16-07-91